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# Optimization of thermoelectric properties in f-electron cage-like intermetallic compounds

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## ABSTRACT

We report on efforts to enhance the thermoelectric properties of the f-electron cage-like intermetallic compounds  $\text{YbT}_2\text{Zn}_{20}$  ( $T = \text{Co, Rh, Ir}$ ), which were previously shown to exhibit a large figure of merit ( $ZT = S^2\sigma T/\kappa$ ) at low temperatures [1]. The factors that define  $ZT$  are the Seebeck coefficient  $S$ , electrical conductivity  $\sigma$ , and thermal conductivity  $\kappa$ , all of which are set by the unusual crystal structure and stoichiometry of these materials: they are ‘electron crystal/phonon glass’ materials [2]. To some extent, all materials with mobile electrons exhibit thermoelectric properties but in most cases it is too weak to be of practical use. These ‘1-2-20’ materials provide counter examples, where the presence of f-electron states near the Fermi energy results in enhanced  $S$ , the abundant conduction electrons produces a large  $\sigma$ , and the large unit cell with a cage like structure minimizes the lattice component of  $\kappa$ . By varying the stoichiometry in these compounds through chemical substitution on the Yb-site, we seek to further optimize these quantities.

## INTRODUCTION

An effective approach to uncovering materials with enhanced thermoelectric properties is to focus on heavy fermion compounds based on Ce and Yb with large or cage-like structures. The expectation that the Seebeck coefficient is high in these compounds stems from the fact that they sometimes feature a large and rapidly changing density of states near the Fermi energy [3, 4]. The crystal structure is also expected to yield a reduced lattice component of the thermal conductivity. We attempted to produce single crystals of  $\text{Yb}_{1-x}\text{Ce}_x\text{Co}_2\text{Zn}_{20}$  and  $\text{Yb}_{1-x}\text{Co}_2\text{Zn}_{20}$ , both of which would increase disorder (lower lattice thermal conductivity) and vary the Kondo lattice energy scale (tune the Seebeck coefficient), by comparison to  $\text{YbCo}_2\text{Zn}_{20}$ . We analyzed their properties using energy-dispersive X-ray spectroscopy, magnetic susceptibility ( $\chi$ ), electrical conductivity ( $\sigma$ ), thermal conductivity ( $\kappa$ ), and Seebeck coefficient ( $S$ ). These measurements show that small amounts of  $\text{Yb} \rightarrow \text{Ce}$  is readily substituted into  $\text{YbCo}_2\text{Zn}_{20}$  and that this varies the total effective magnetic moment while preserving the Curie-Weiss behavior. In the intermediate x-range the stoichiometry varies dramatically from the 1-2-20 ratio, suggesting the formation of an unexpected phase.

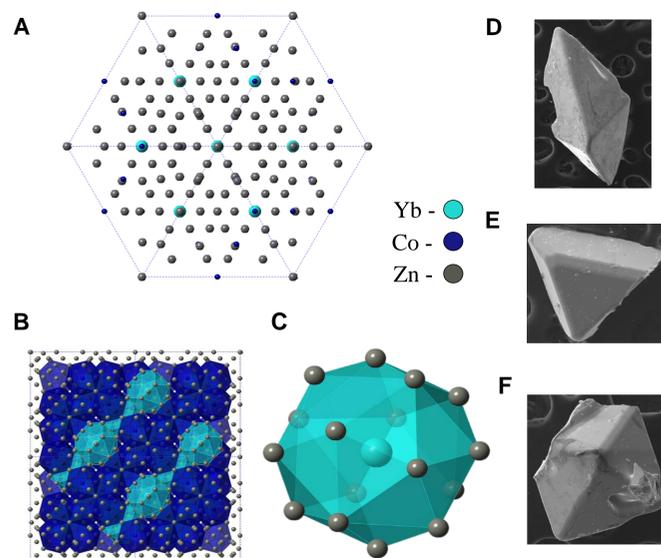
## SAMPLE SYNTHESIS

$\text{Yb}_{1-x}\text{Ce}_x\text{Co}_2\text{Zn}_{20}$  and  $\text{Yb}_{1-x}\text{Co}_2\text{Zn}_{20}$  crystals were synthesized using the molten metallic flux growth technique. The starting materials were Yb chunks (99.9%, Ames Labs), Ce rods (99.9%, Ames Labs), Co ingots (99.99%, Alfa Aesar), and Zn shot (99.999%, Alfa Aesar). Elements were combined in the atomic ratios  $\text{Yb}:\text{Co}:\text{Zn} = 0.1:2:60$ ,  $0.05:2:60$ ,  $0.025:2:60$  and  $\text{Yb}:\text{Ce}:\text{Co}:\text{Zn} = 0.75:0.25:2:60$ ,  $0.5:0.5:2:60$ ,  $0.25:0.75:2:60$ ,  $0:1:2:60$ . The elements were loaded into 2-ml alumina Canfield crucibles and sealed under vacuum in quartz tubes. The quartz tubes were placed in a resistive box furnace then heated to  $300^\circ\text{C}$  at a rate of  $50^\circ\text{C}/\text{hour}$ , held at  $300^\circ\text{C}$  for 2 hours, heated to  $1050^\circ\text{C}$  at a rate of  $50^\circ\text{C}/\text{hour}$ , held at  $1050^\circ\text{C}$  for 72 hours, then cooled to  $700^\circ\text{C}$  at a rate of  $4^\circ\text{C}/\text{hour}$ , held at  $700^\circ\text{C}$  for 48 hours. At this temperature, the samples were removed from the furnace and centrifuged to separate the remaining flux from the precipitated crystals.

## FUTURE WORK

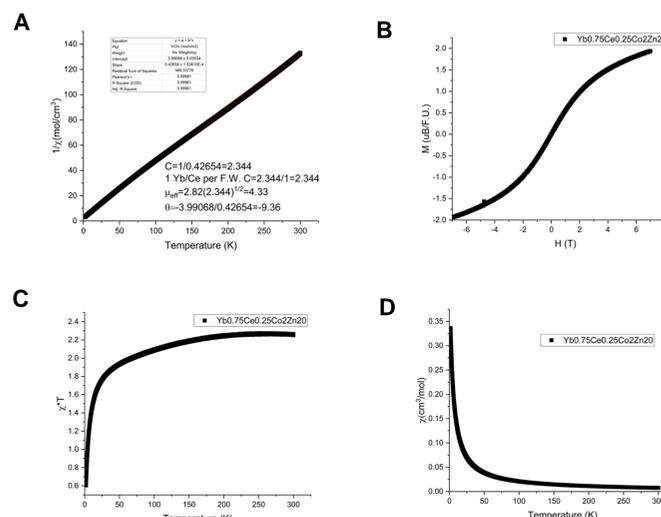
- Measure the temperature-dependent power factor and thermoelectric figure of merit for BS-10 and BS-9 to determine whether the disorder has caused enhanced thermoelectric properties.
- Further characterize the BS-4, BS-5, and BS-6 compounds that provided an unexpected phase.

## CRYSTAL STRUCTURE



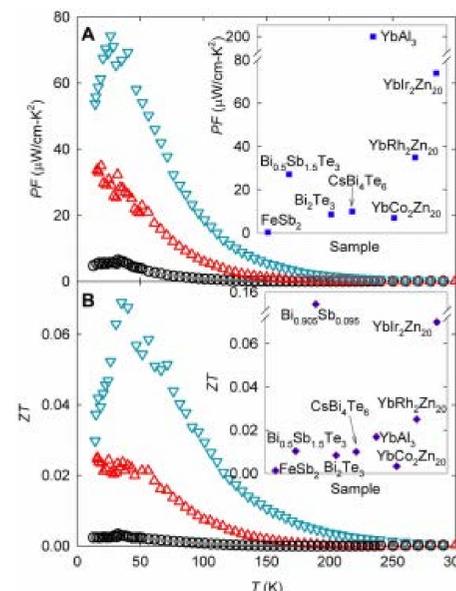
**Figure 1:** (A) [111] directional view of one unit cell of  $\text{YbCo}_2\text{Zn}_{20}$ . (B) [100] directional view of polyhedron unit cells. (C) Single ytterbium ion surrounded by a zinc cage-like structure. The images of the single crystals of  $\text{YbCo}_2\text{Zn}_{20}$  (D), (E), and (F) were captured using a Scanning Electron Microscopy (SEM).

## MAGNETIC SUSCEPTIBILITY



**Figure 3:** Temperature and field dependent magnetization measurements performed using a Quantum Design VSM Magnetic Property Measurement System (MPMS). (A) represents the inverse magnetic susceptibility  $1/\chi$  vs. temperature  $T$ , where the straight line is a Curie-Weiss fit to the data. (B) represents magnetization  $M$  vs field  $H$ . (C) represents the product of magnetic susceptibility and temperature  $\chi T$  vs  $T$ . (D)  $\chi$  vs  $T$ .

## THERMOELECTRICS



**Figure 2:** Temperature-dependent (A) power factor and (B) thermoelectric figure of merit from parent compounds [1].  $\text{YbCo}_2\text{Zn}_{20}$  (circle),  $\text{YbRh}_2\text{Zn}_{20}$  (up-triangle), and  $\text{YbIr}_2\text{Zn}_{20}$  (down-triangle).

## CHEMICAL ANALYSIS

Sample #	Target Composition %				EDS Composition %			
	Yb	Ce	Co	Zn	Yb	Ce	Co	Zn
BS-3	3.26	1.09	8.7	86.96	3.54	1.09	9.03	86.12
BS-4	2.17	2.17	8.7	86.96	1.04	9.925	2.2	86.84
BS-5	1.09	3.26	8.7	86.96	0.56	10.03	2.28	87.14
BS-6	0	4.38	8.7	86.96	0.58	9.87	2.25	87.3
BS-10	3.26	1.09	8.7	86.96	4.01	0.79	9.03	86.17

Sample #	Target Composition %			EDS Composition %		
	Yb	Co	Zn	Yb	Co	Zn
BS-7	4.24	8.7	86.96	4.85	9.24	85.92
BS-8	4.13	8.7	86.96	4.79	9.13	86.09
BS-9	3.91	8.7	86.96	4.86	9.04	86.11

**Figure 4:** Energy-dispersive X-ray spectroscopy (EDS) used to determine chemical composition. (Table A) represents the comparison between the target composition percentage versus the measured EDS composition percentage in  $\text{Yb}_{1-x}\text{Ce}_x\text{Co}_2\text{Zn}_{20}$ . (Table B) represents the comparison between the target composition percentage versus the measured EDS composition percentage in  $\text{Yb}_{1-x}\text{Co}_2\text{Zn}_{20}$ .

## PHENOMENOLOGICAL THEORY

Thermoelectric Figure of Merit

$$ZT = \frac{S^2 \sigma}{\kappa} T$$

- $S$  - Seebeck coefficient
- $T$  - absolute temperature
- $\kappa$  - thermal conductivity
- $\sigma$  - electrical conductivity

Seebeck coefficient

$$S = \frac{\Delta V}{\Delta T}$$

- $S$  - Seebeck coefficient
- $V$  - electric potential
- $T$  - temperature

Power Factor

$$PF = \frac{S^2}{\rho}$$

- $S$  - Seebeck coefficient
- $\rho$  - electrical resistivity

## CONCLUSIONS

- Based on EDS data, the target concentrations of Yb and Ce were successfully substituted for the  $\text{Yb}_{0.75}\text{Ce}_{0.25}\text{Co}_2\text{Zn}_{20}$  compound.
- Magnetic susceptibility data from  $\text{Yb}_{0.75}\text{Ce}_{0.25}\text{Co}_2\text{Zn}_{20}$  proves that the total effective magnetic moment value varies from the Yb standard of  $4.5\mu_B$  to the measured  $4.33\mu_B$ , while retaining the Curie-Weiss behavior.
- The stoichiometry alters dramatically from the target compositions at ratios other than  $\text{Yb}_{0.75}\text{Ce}_{0.25}\text{Co}_2\text{Zn}_{20}$ , this indicates the formation of an unexpected phase in BS-4, BS-5, and BS-6.

## REFERENCES

- [1] K. Wei, J. N. Neu, Y. Lai, K.-W. Chen, D. Hobbis, G. S. Nolas, D. E. Graf, T. Siegrist, R. E. Baumbach, Enhanced thermoelectric performance of heavy-fermion compounds  $\text{YbTM}_2\text{Zn}_{20}$  ( $\text{TM} = \text{Co, Rh, Ir}$ ) at low temperatures. *Sci. Adv.* 5, eaaw6183 (2019).
- [2] G. A. Slack, in *CRC Handbook of Thermoelectrics*, D. M. Rowe, ED. (CRC, 1995), 407 pp.
- [3] B. C. Sales, Novel thermoelectric materials. *Curr. Opin. Solid State Mater. Sci.* 2, 284-289 (1997).
- [4] G. D. Mahan, J. O. Sofo, The Best Thermoelectric. *Proc. Natl. Acad. Sci. U.S.A.* 93, 7436-7439 (1996)

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